



The Effects of Post-Processing on Mechanical Properties and Corrosion Behavior of AM 316L Stainless Steel

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US Naval Research Laboratory
Codes 6350 and 6130

Supported by NRL and the NAVSEA Technology Office
Cross Platform Systems Development (CPSD) Program

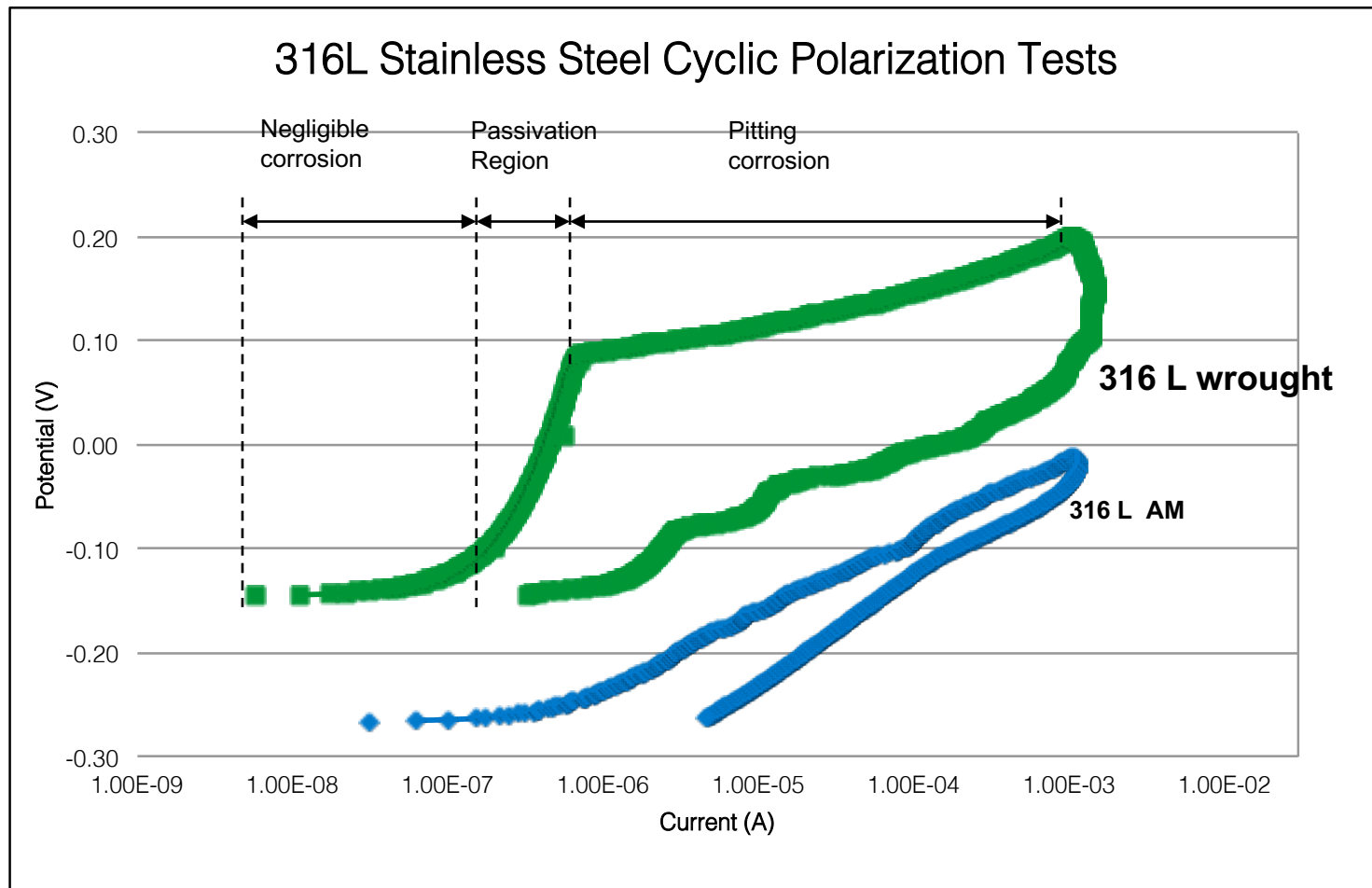
Objectives: Systematically determine the microstructure, corrosion behavior, and mechanical properties of AM 316L stainless steel in the as-built and post-processed conditions

Approach: Take advantage of the outstanding capabilities and expertise in microstructural characterization and corrosion behavior at NRL:

- Advanced 2D and 3D microscopy techniques
- Corrosion testing
- Mechanical property testing

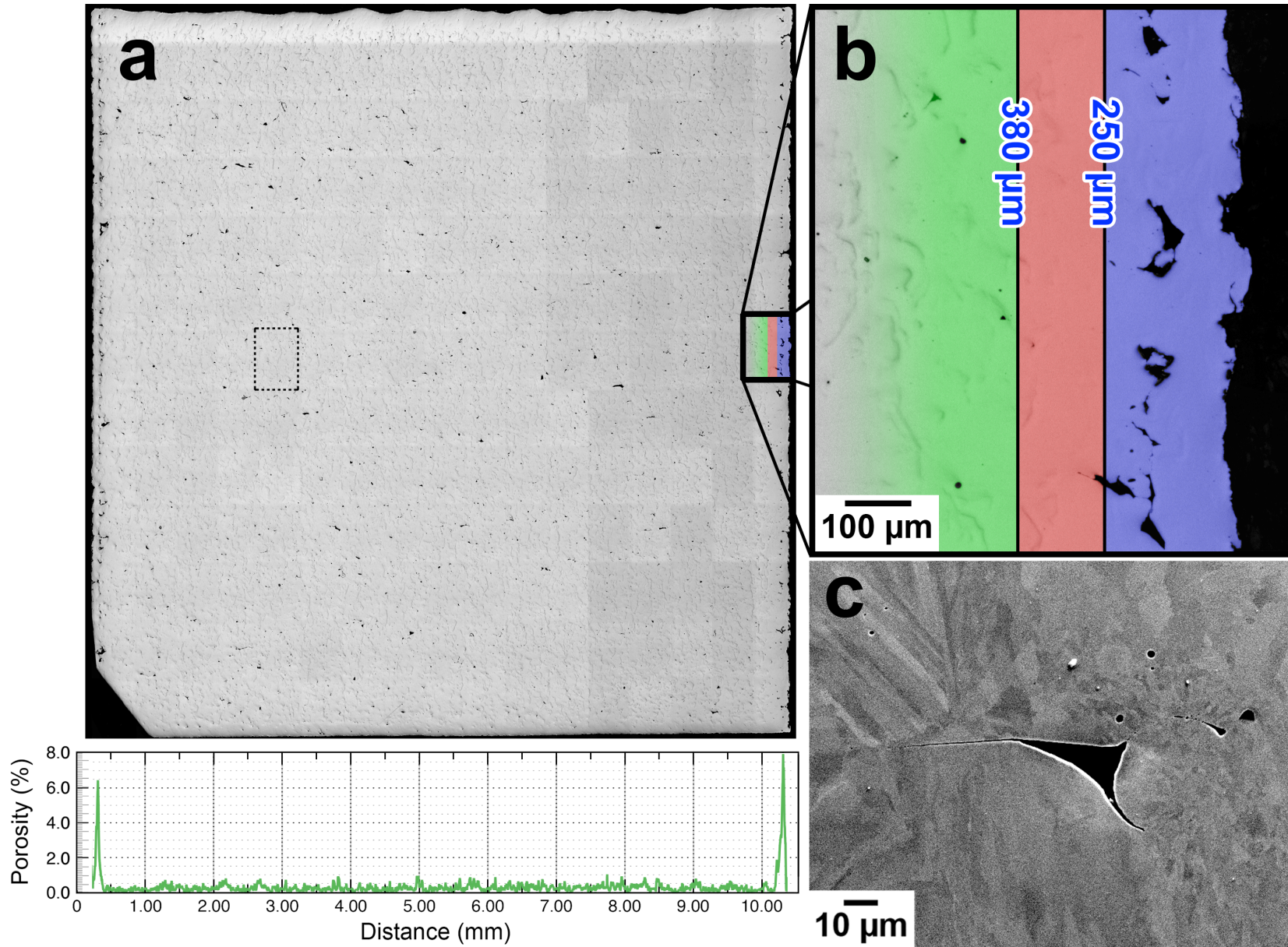
316L AM Steel is NOT Stainless

- As-built 316L sample does not exhibit any passivity
- AM sample corrodes three orders of magnitude faster than wrought sample at -100mV



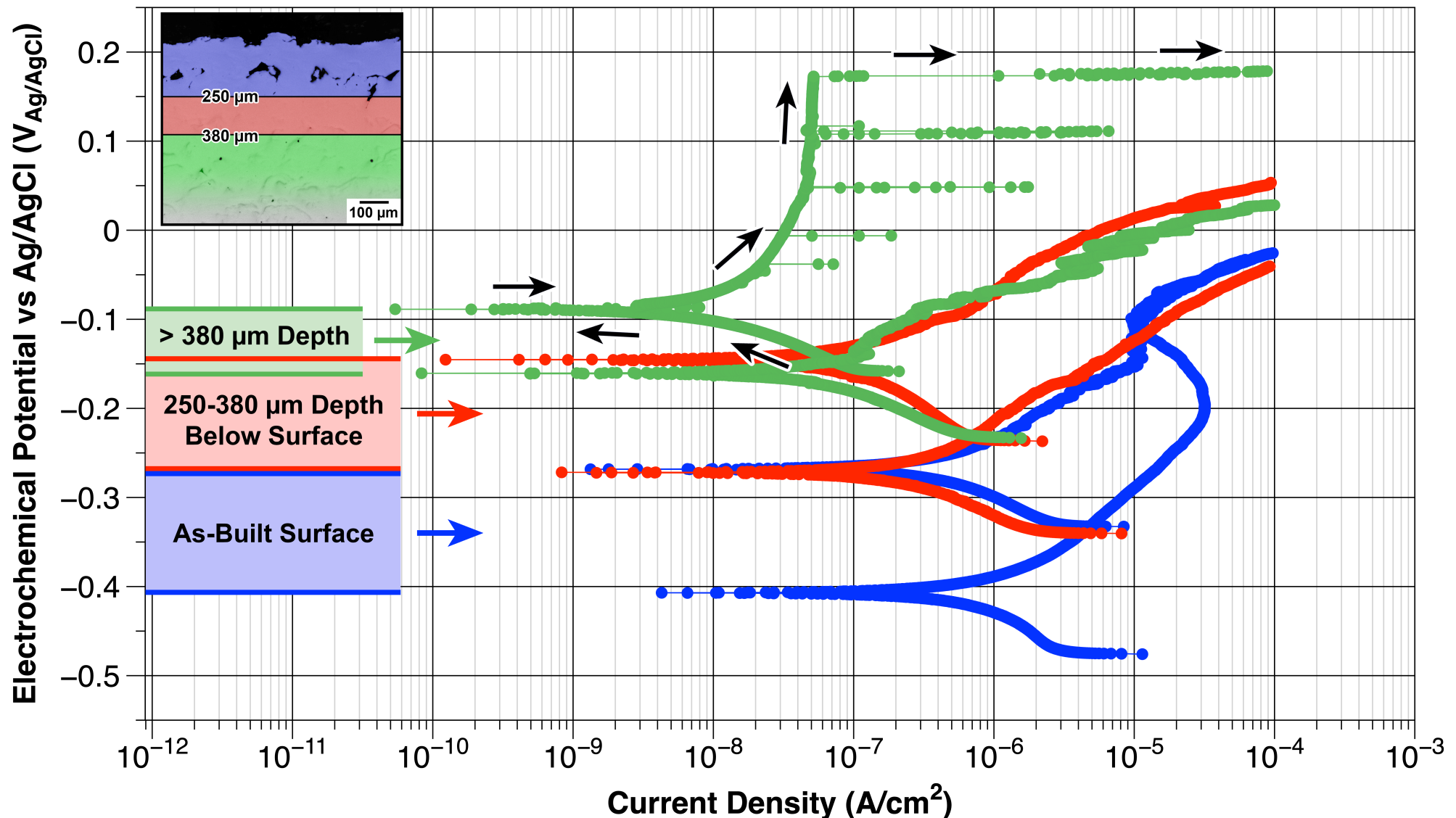
Porosity in As-Built Material

Both LOF and gas porosity present
Enhanced porosity between fill and border scans

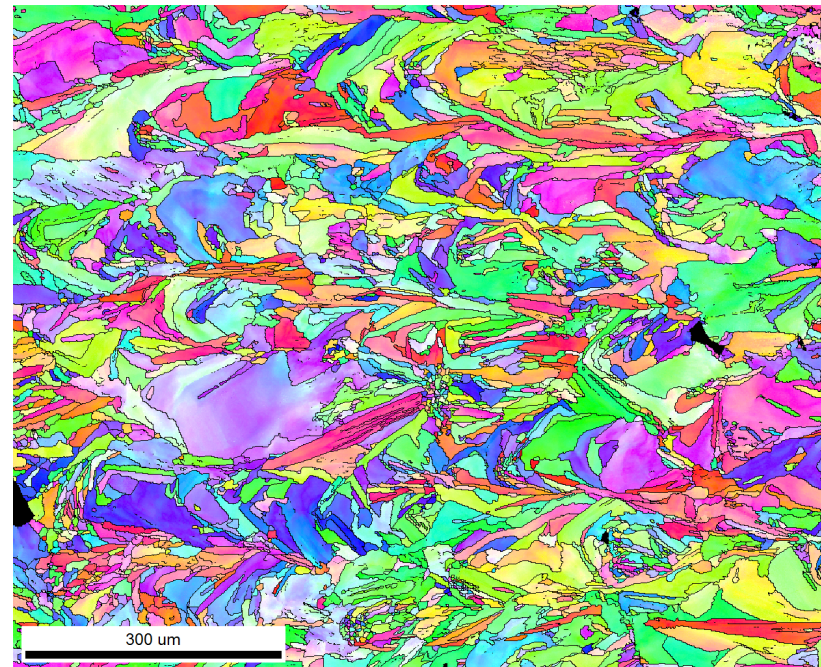
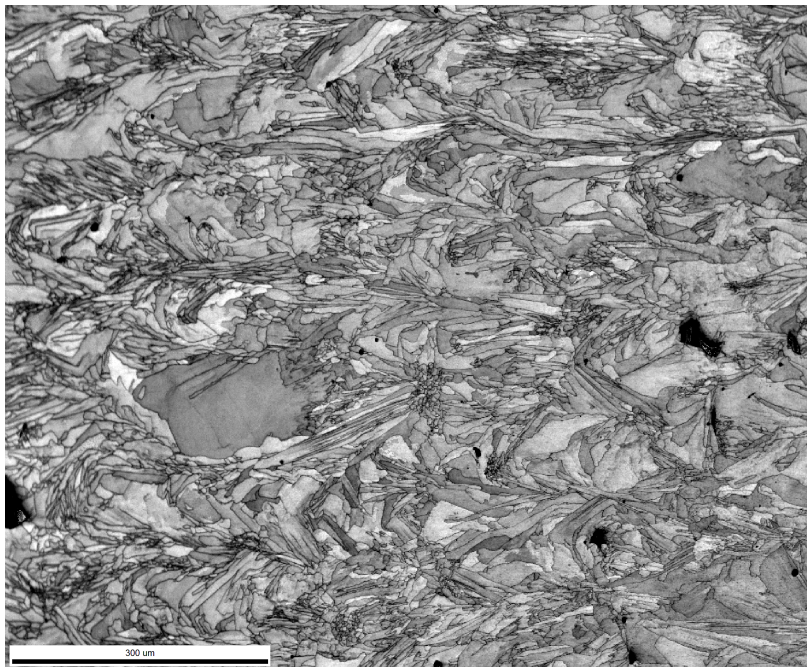
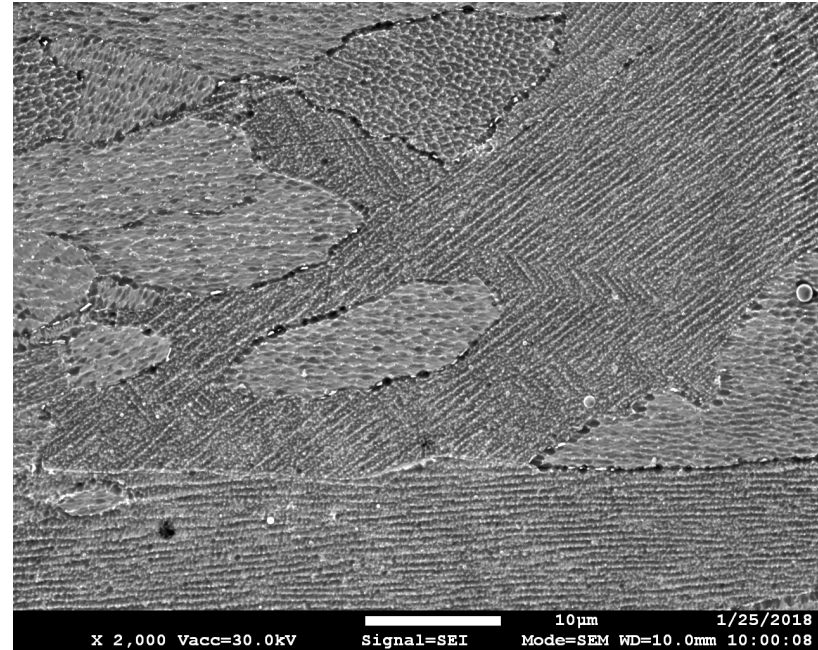
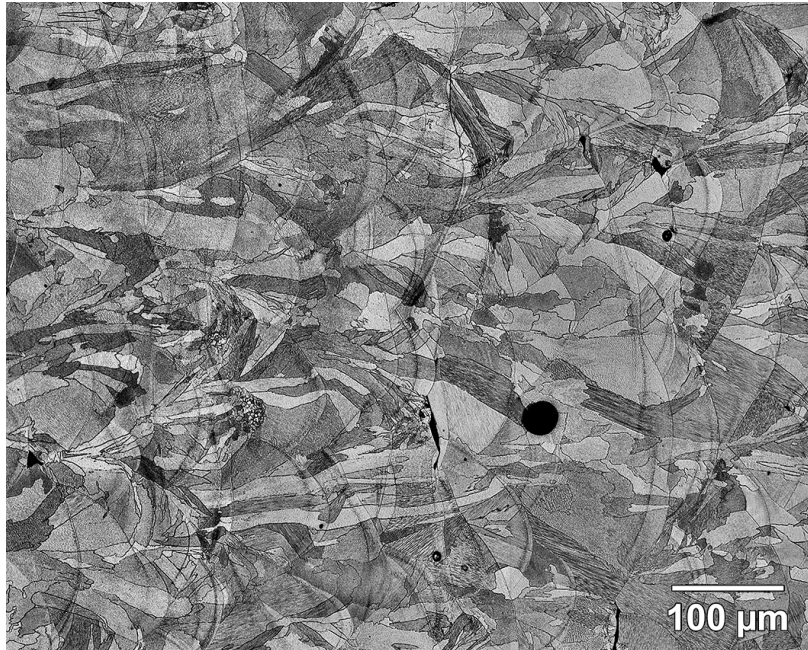


Depth Effects of Corrosion

- No passivity at the as-built surface
- Passivity improves with distance from surface



Standard L-PBF Appearance



- **As-Received Sample:**

- 316L Stainless Steel
- EOS M270
- Stress Relief: 790 °C, 1 h

- **Additional Heat Treatments:**

500 °C, 1h

1100 °C, 1h

700 °C, 1h

1200 °C, 1h

800 °C, 1h

1300 °C, 1h

900 °C, 1h

1300 °C, 15h

1000 °C, 1h

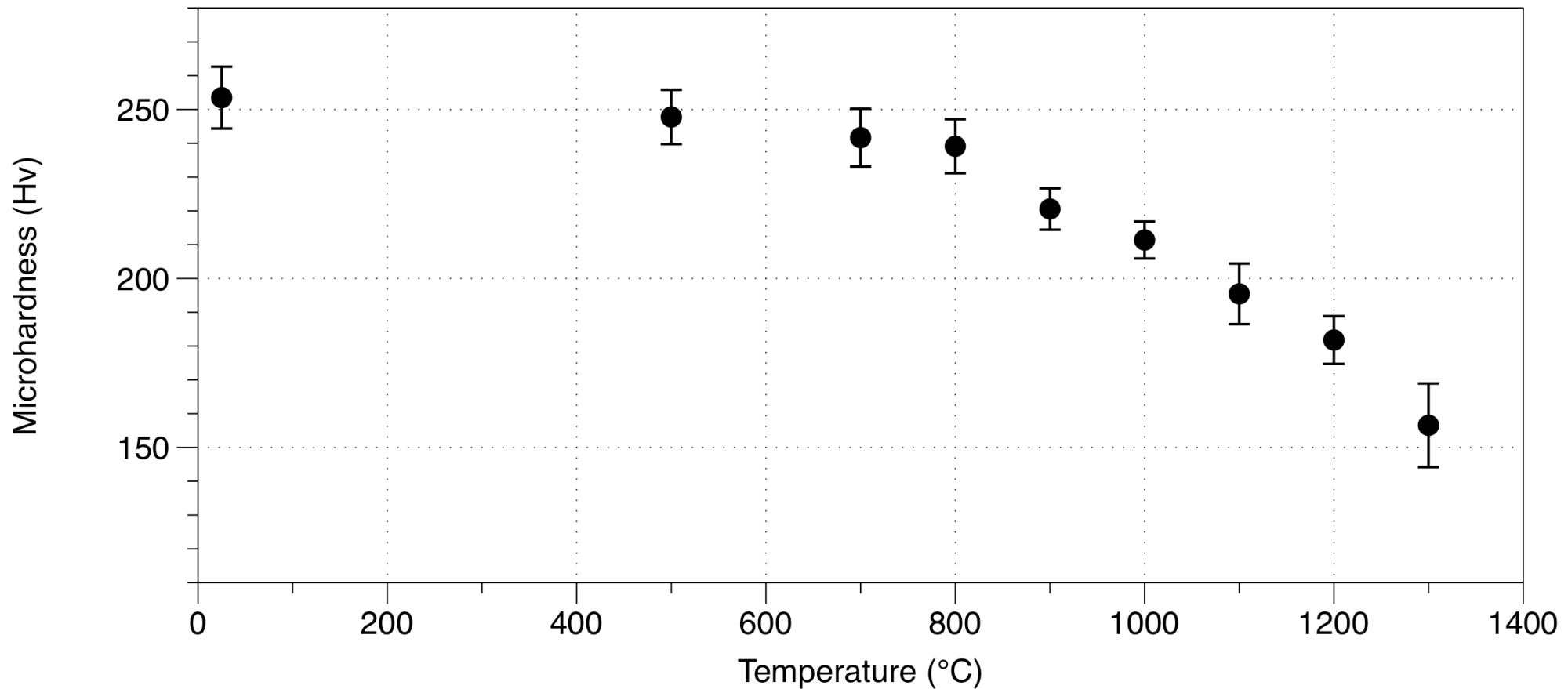
- **HIP Treatments—15 ksi (100 Mpa):**

1000 °C, 3h

1100 °C, 3h

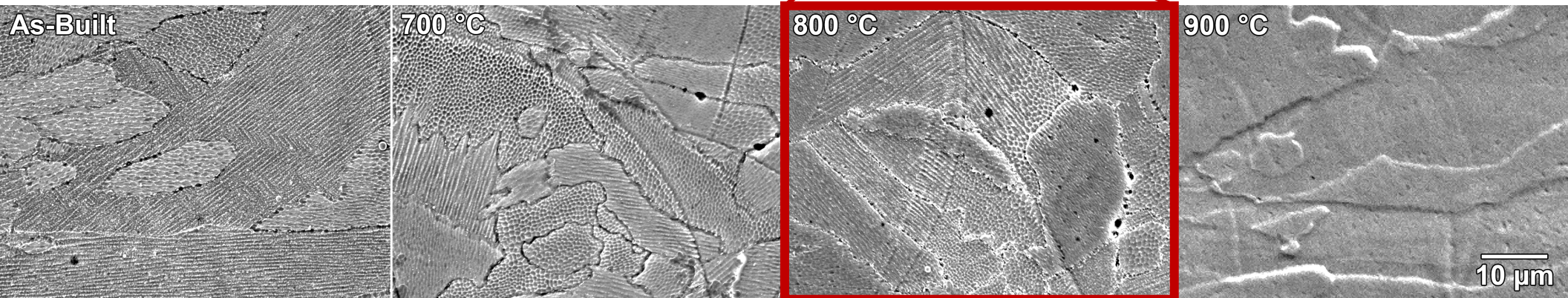
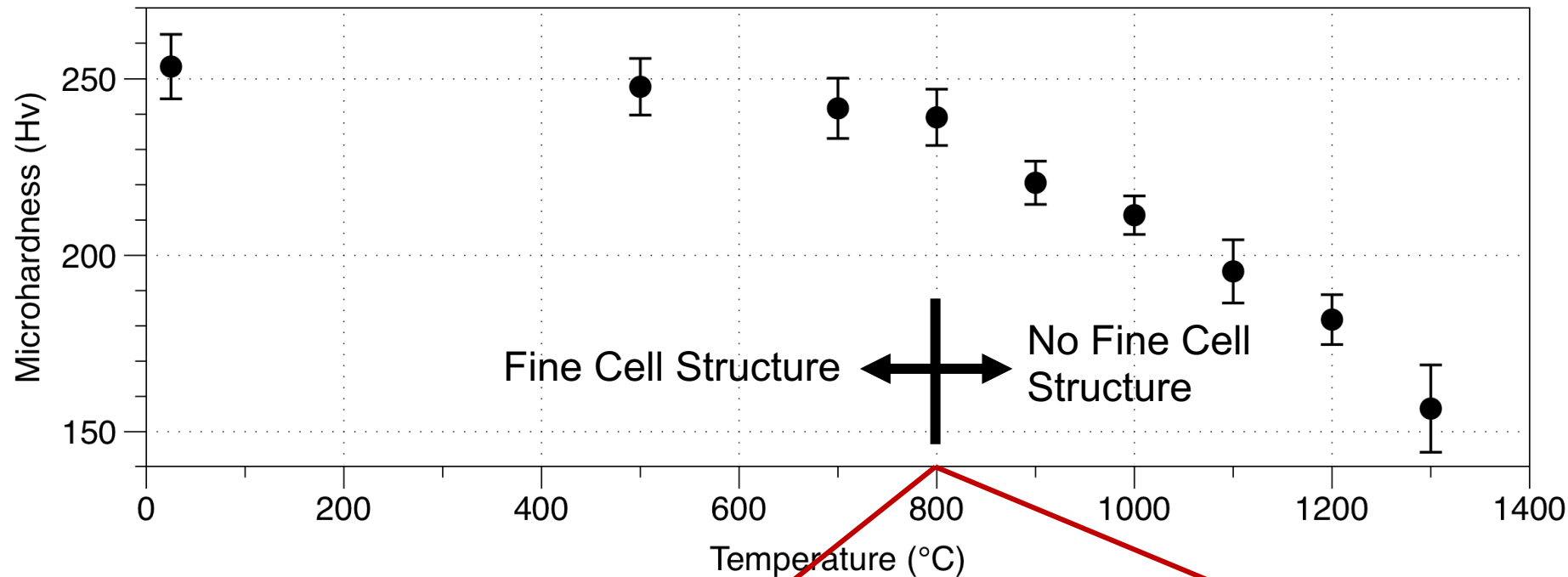
1200 °C, 3h

Microhardness Variations



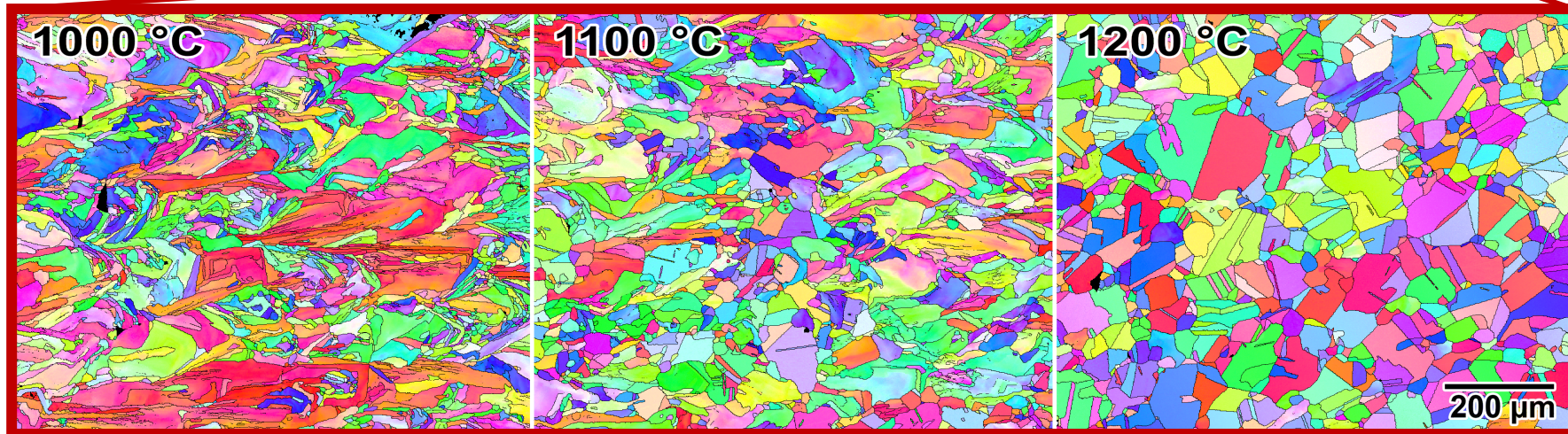
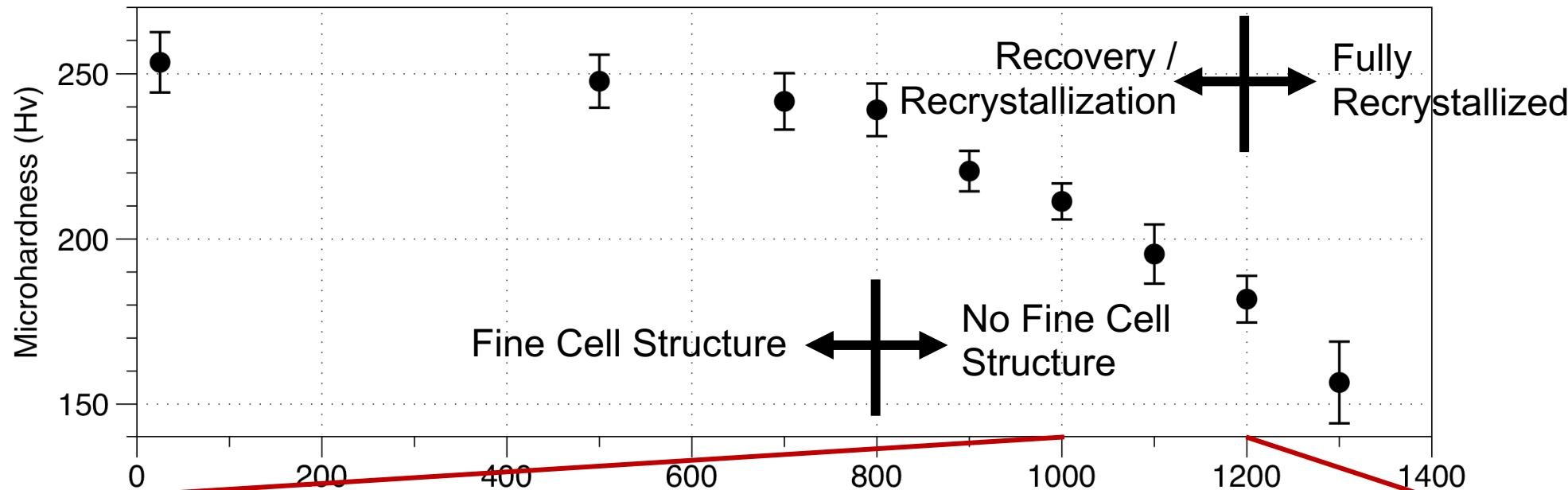
- Heat treat for 1 hour at various Temps
 - Slight decrease up to 800 °C
 - Steeper decrease above 800 °C

Changes in Cell Structure



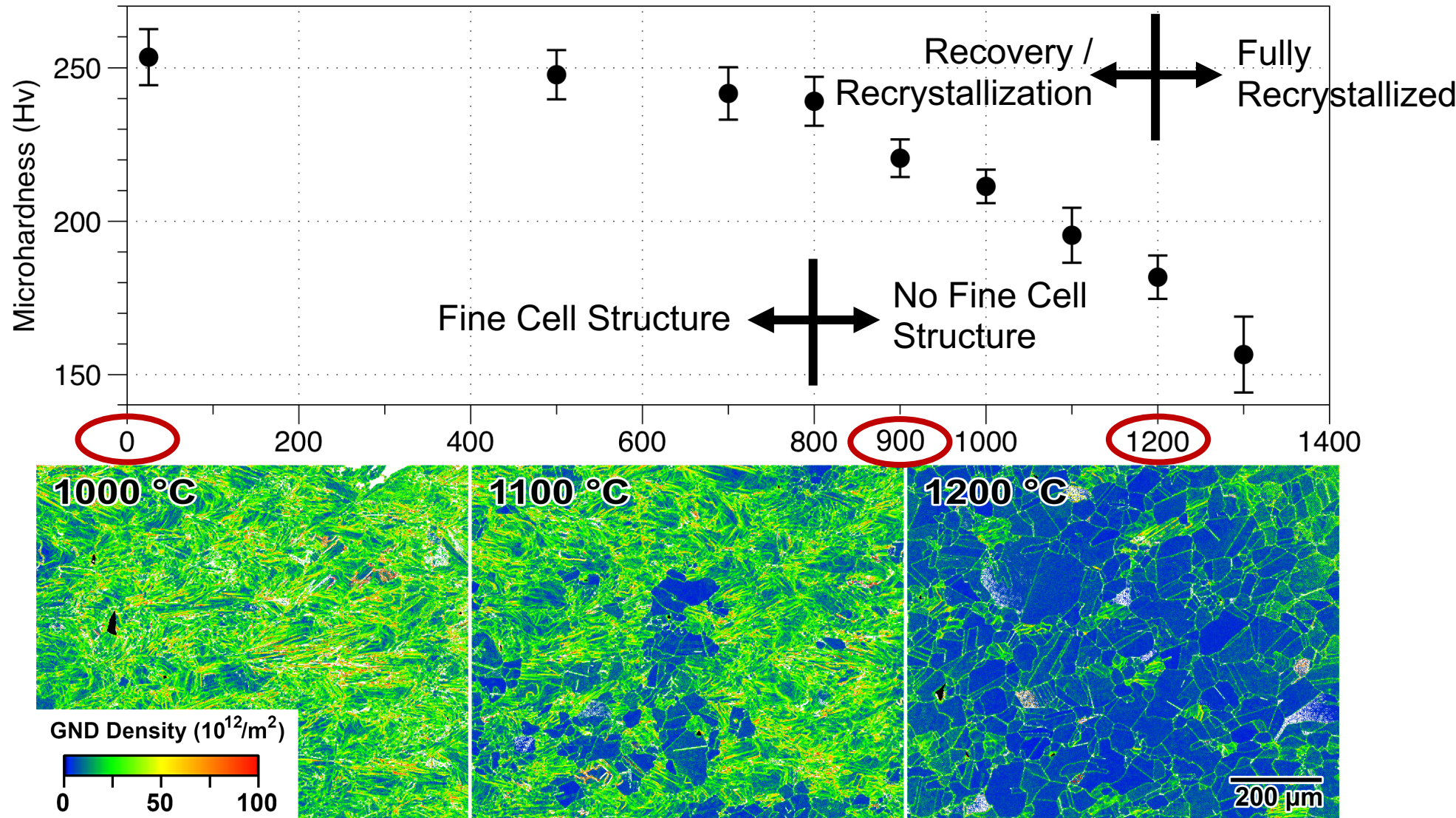
- Cell structure evident below 800 °C
- Associated with higher strength and better corrosion resistance
- Completely absent at 900 °C and above

Grain Evolution



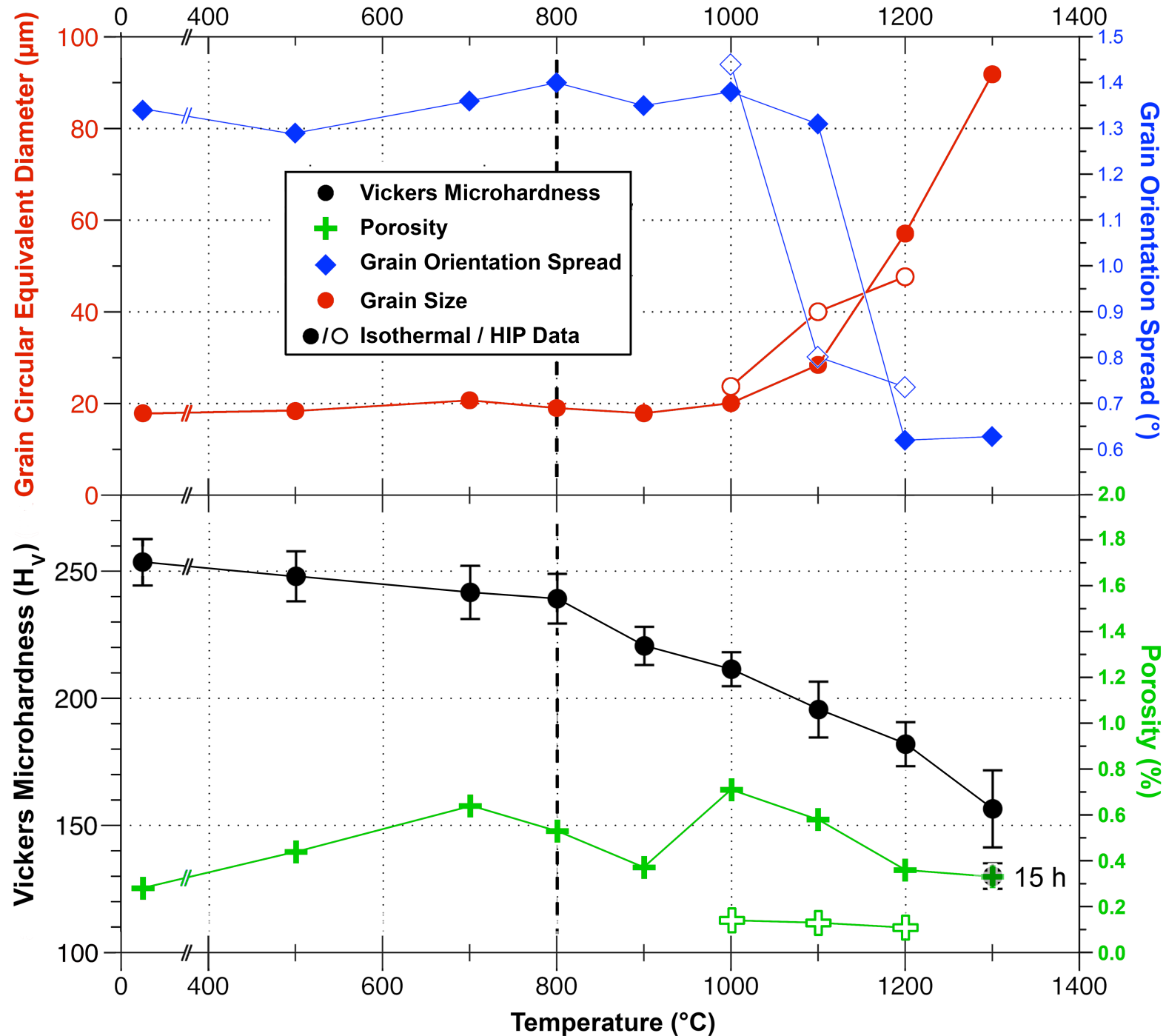
- Recovery / Recrystallization occurs from 1000-1200 °C
- Difficult to discern from IPF maps

Recrystallization

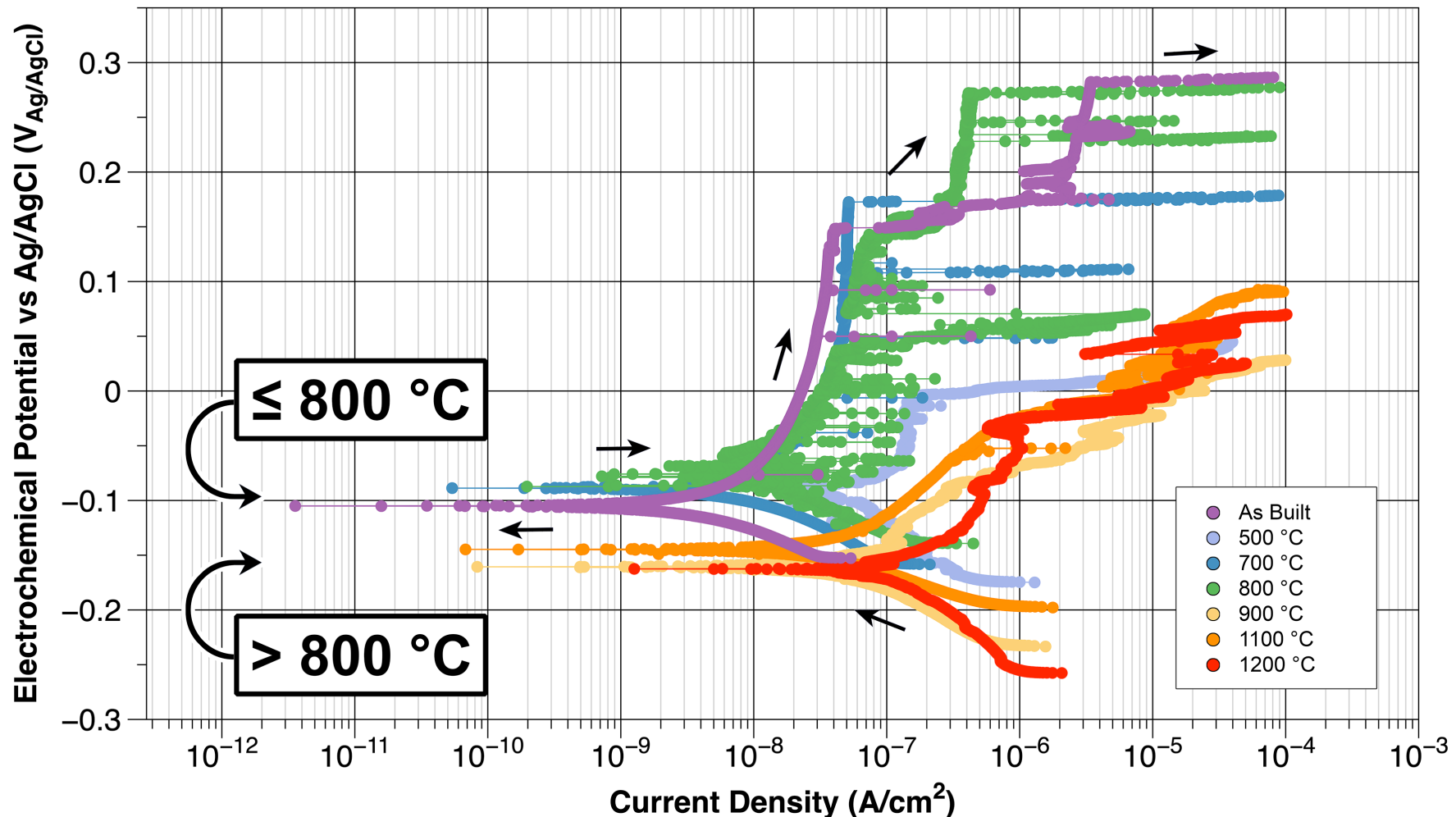


- Recovery / Recrystallization occurs from 1000-1200 °C
- More evident in Geometrically Necessary Dislocation maps
- **Three characteristic temperature regimes**

Evolution with Temperature

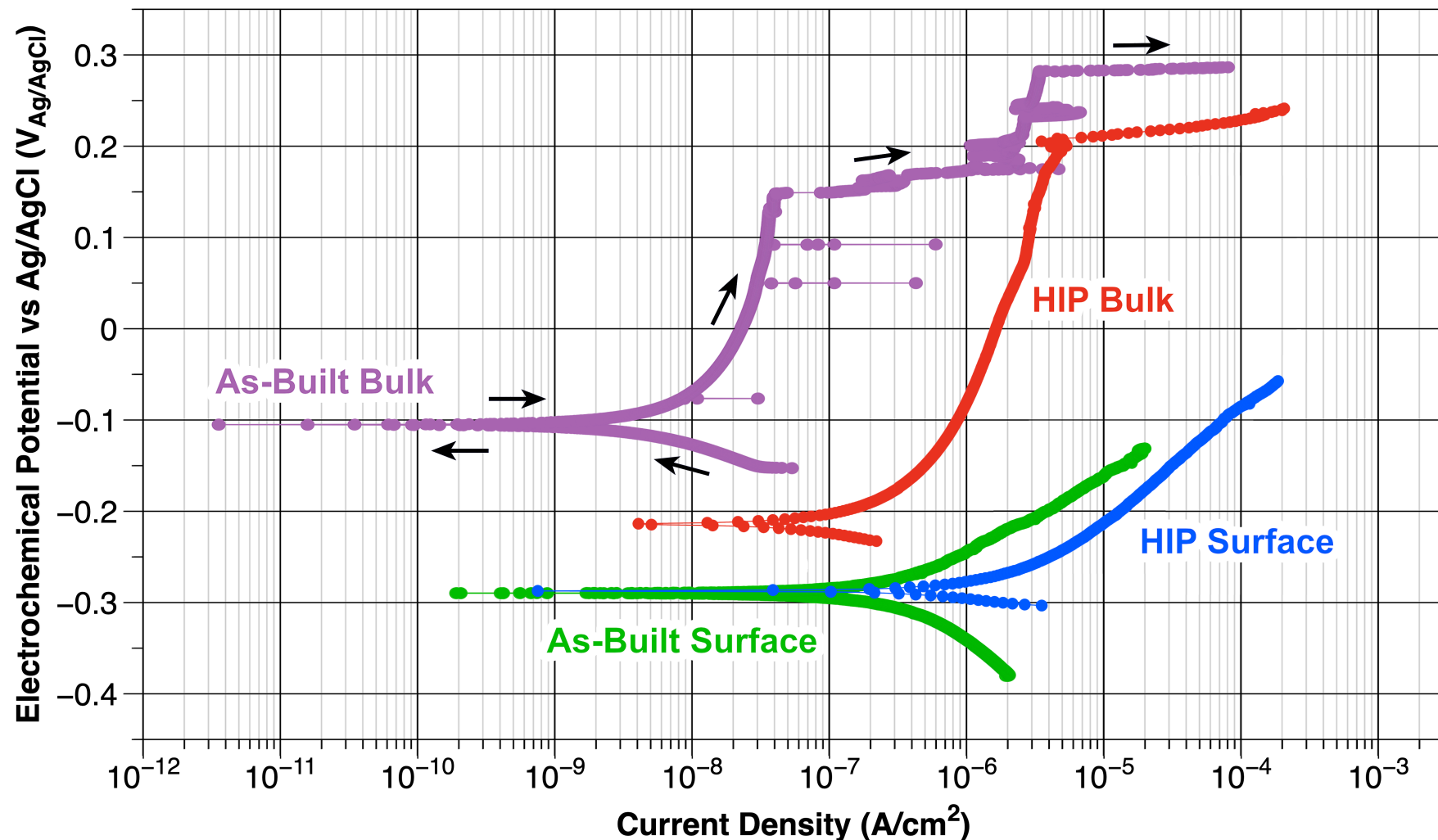


Corrosion of Bulk AM 316L



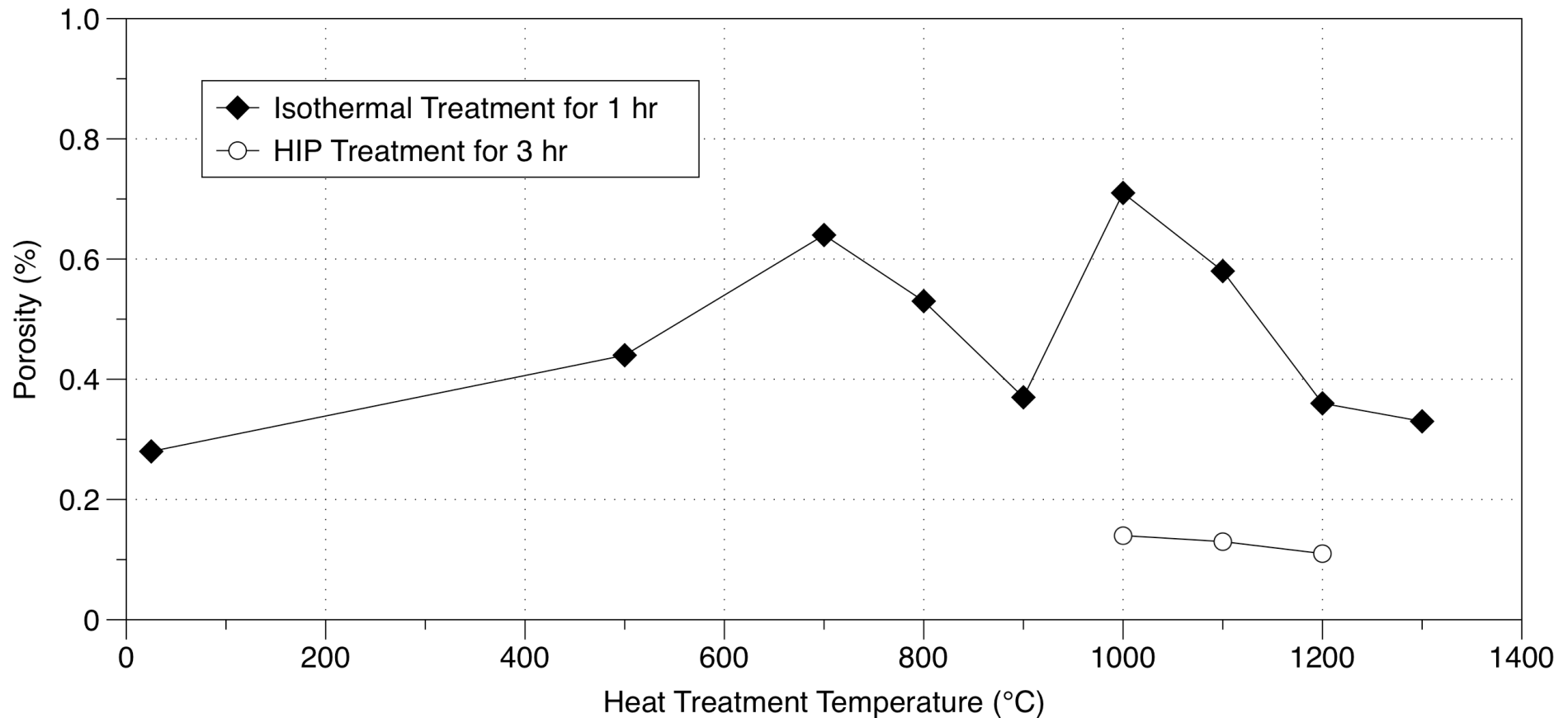
- Passive behavior of build interior up to 800 °C
- Corrosion is enhanced / passivity lost above 800 °C

Corrosion of Bulk HIP AM 316L



- Enhanced corrosion rate after HIP
- Passive / non-passive behavior retained

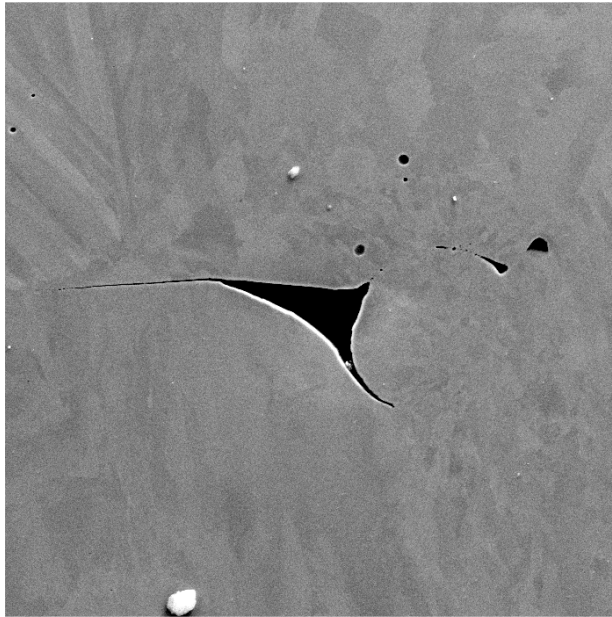
Effect of T on Porosity Fraction



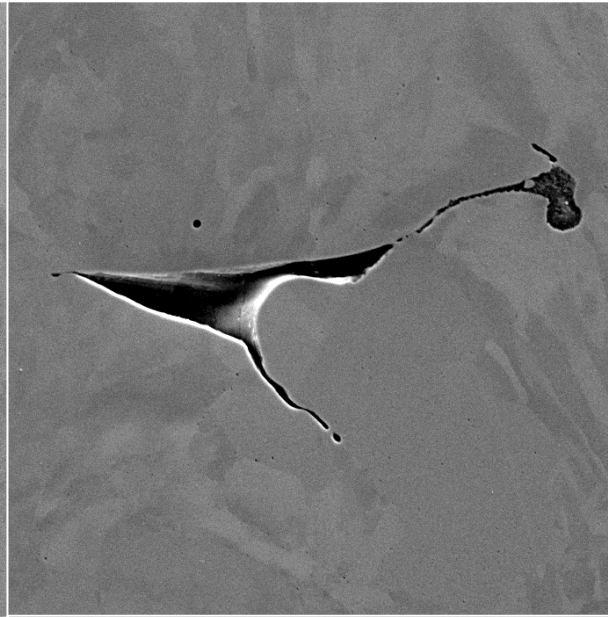
- Heat treatments alone do not change pore volume fraction
- HIP treatments reduce pore volume fraction by ~2/3

Effect of HT on Pore Morphology

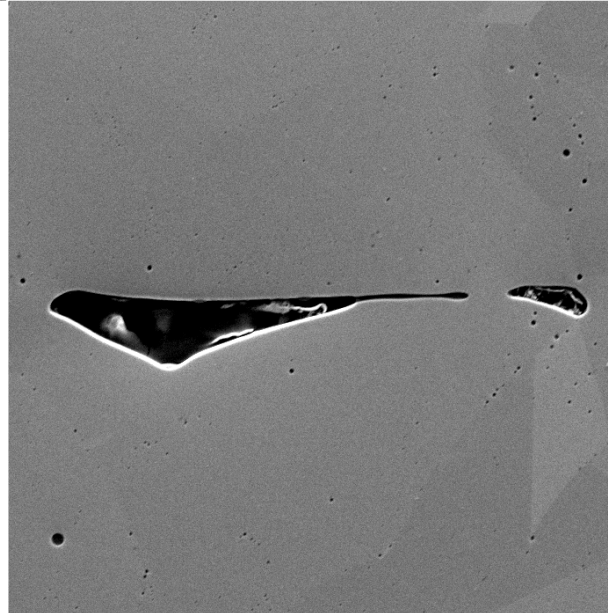
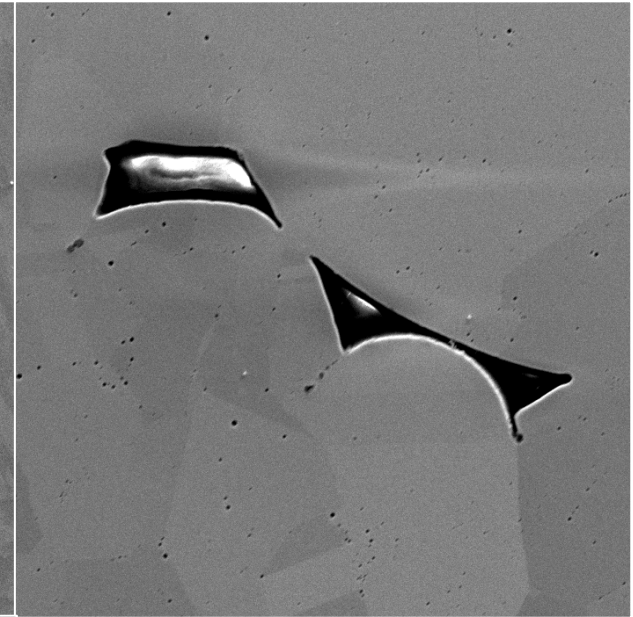
As-received



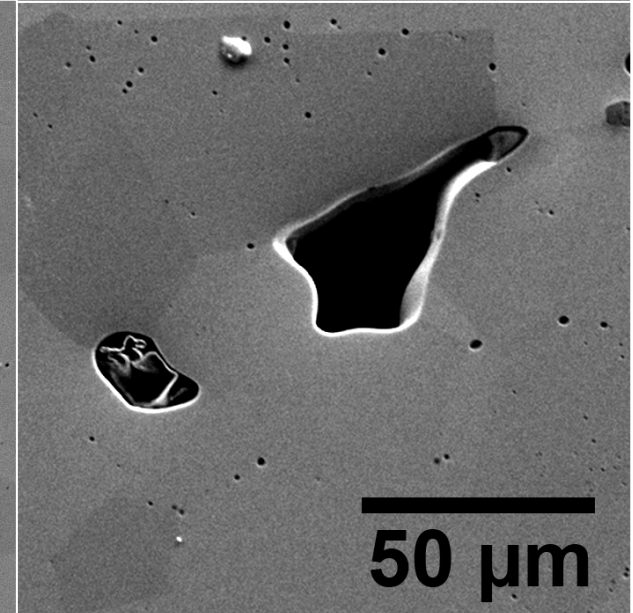
1100 °C, 1 hr



1200 °C, 1 hr



1300 °C, 1 hr



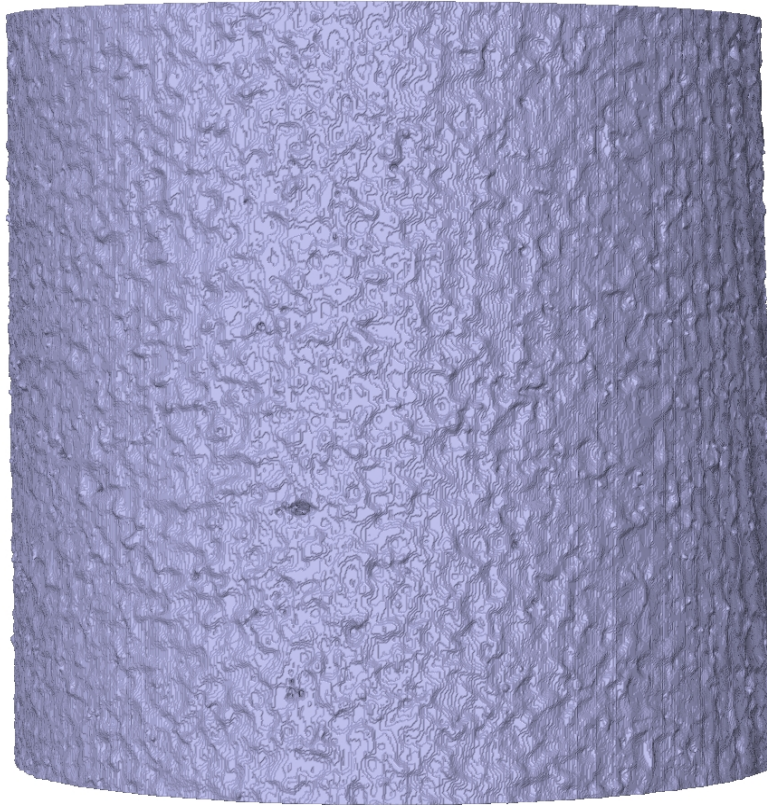
1300 °C, 15 hr

50 μm

- No effects at <1000 °C
- Initial rounding of pore corners at 1100 °C
- No gross changes in pore shape <1300 °C
- Follows recrystallization behavior

Effects of HIP on Porosity

As-Built



1.0 mm

HIP



0.75 mm

LOF pores removed by HIP; small pores remain

Further analysis of four characteristic structures

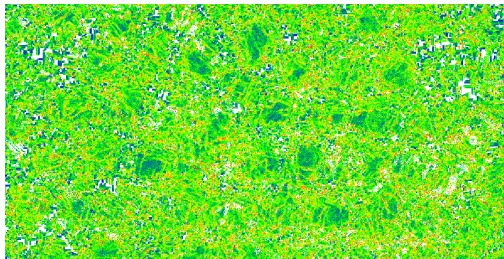
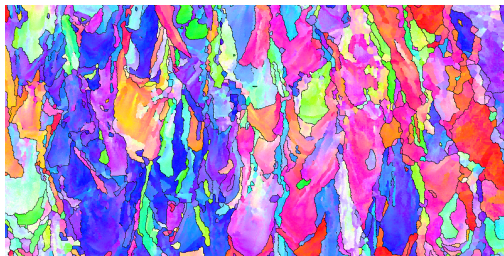
- EOS M290 at JHU-APL
- EOS StainlessSteel 316L
- **Porosity $<0.1\%$**

Parameters

- Power: 195 W
- Laser Speed: 1083 mm/s
- Layer thickness: 0.02 mm
- Hatch Distance: 0.09 mm
- Stripe width: 5 mm
- Stripe overlap: 0.12 mm

Characteristic Microstructures

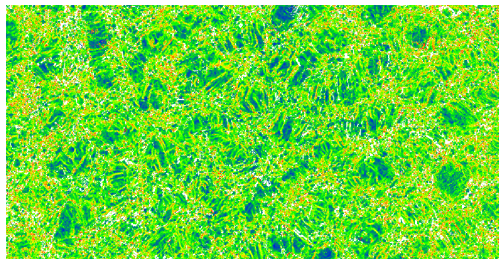
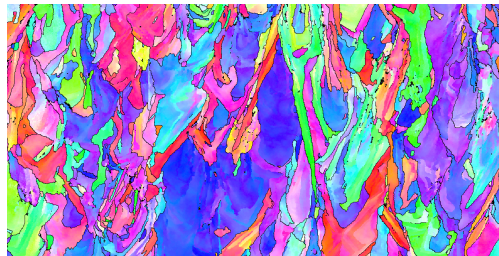
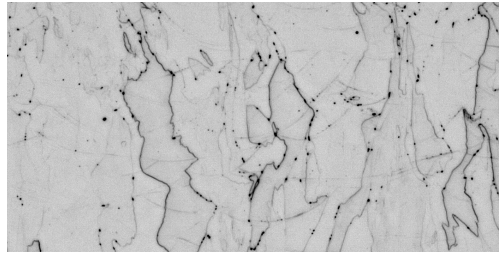
As-Built



As-Built

Fine cell structure
As-built grain
structure
Few precipitates

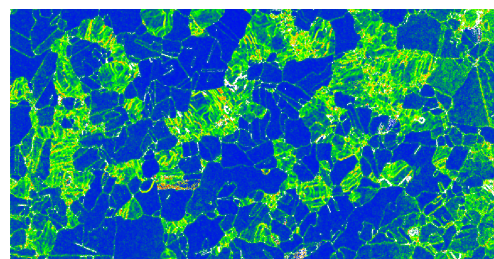
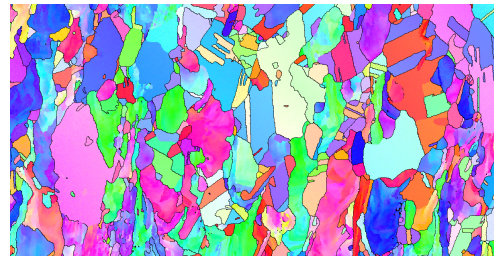
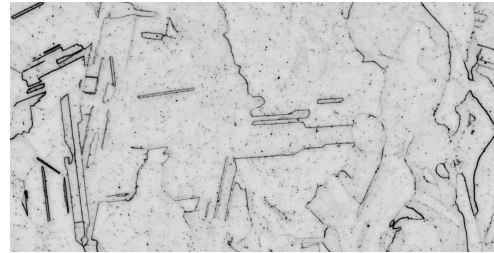
Annealed



900 °C, 1h

No cell structure
As-built grain
structure
Few precipitates

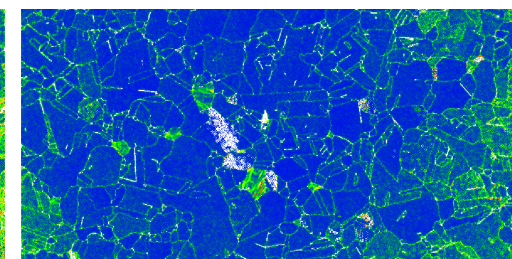
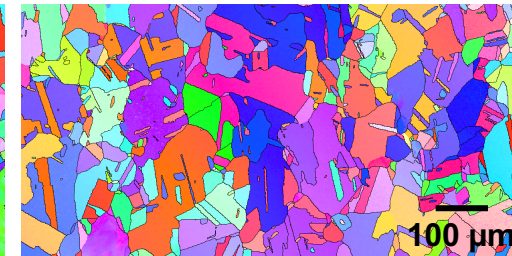
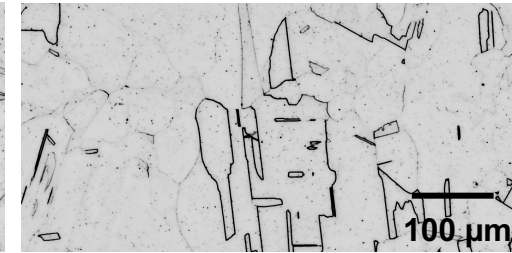
Recrystallized



1200 °C, 1h

No cell structure
**70% recrystallized
grain structure**
Fine precipitates

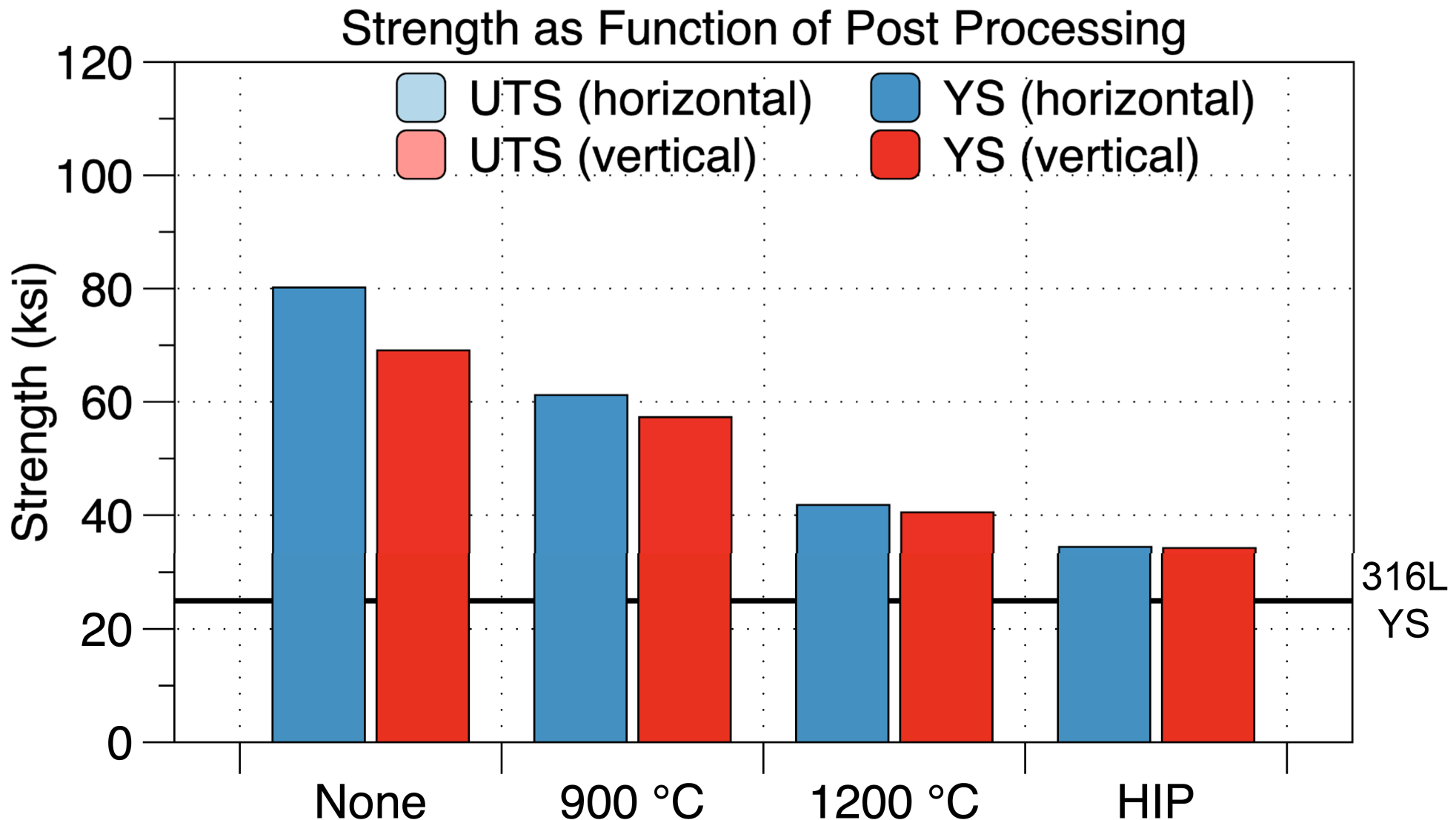
HIPed



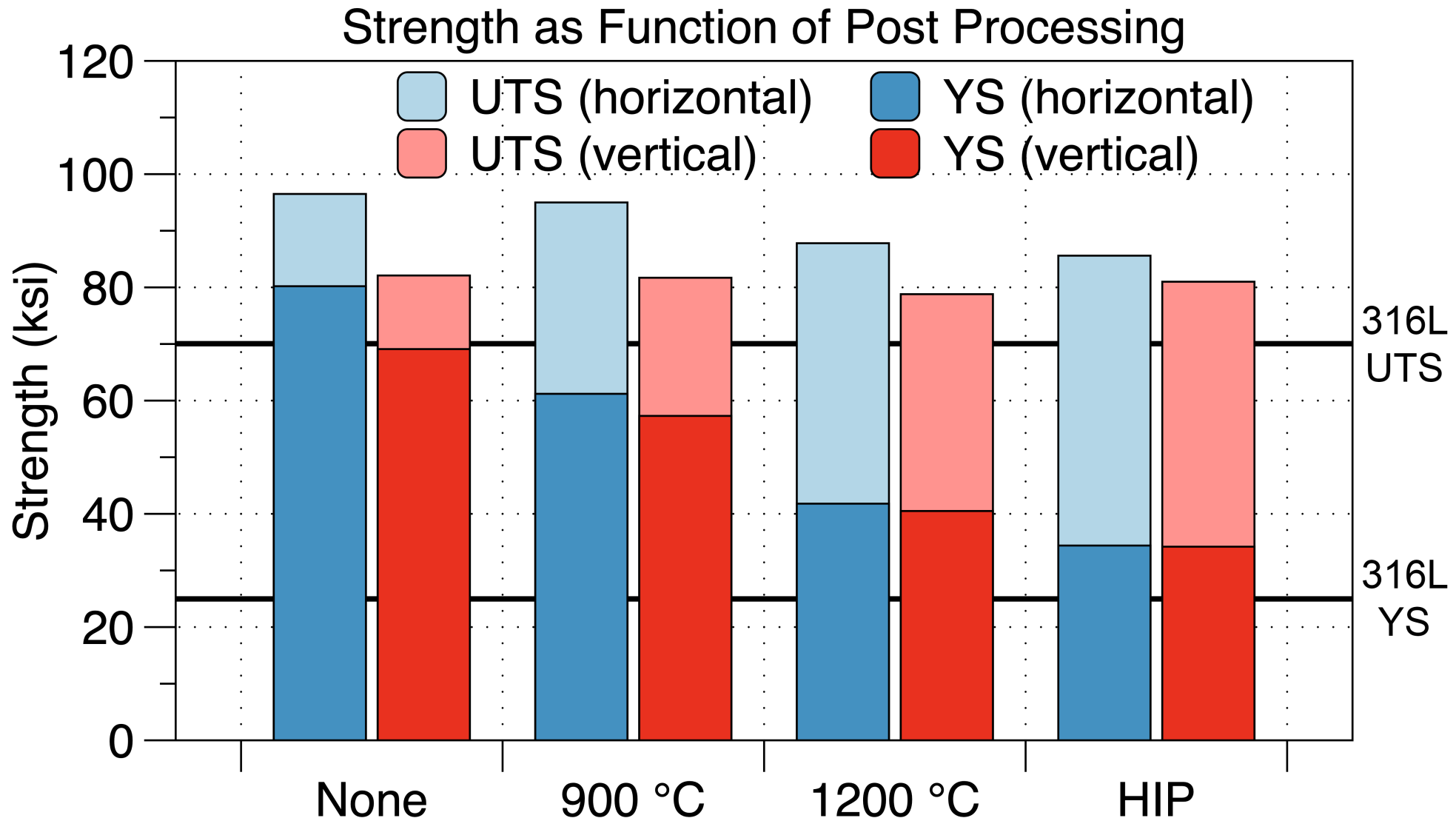
HIP: 1204 °C, 4h

No cell structure
**Fully recrystallized
grain structure**
Fine precipitates

- YS decreases with processing temperature
- All samples exceed 316L specifications

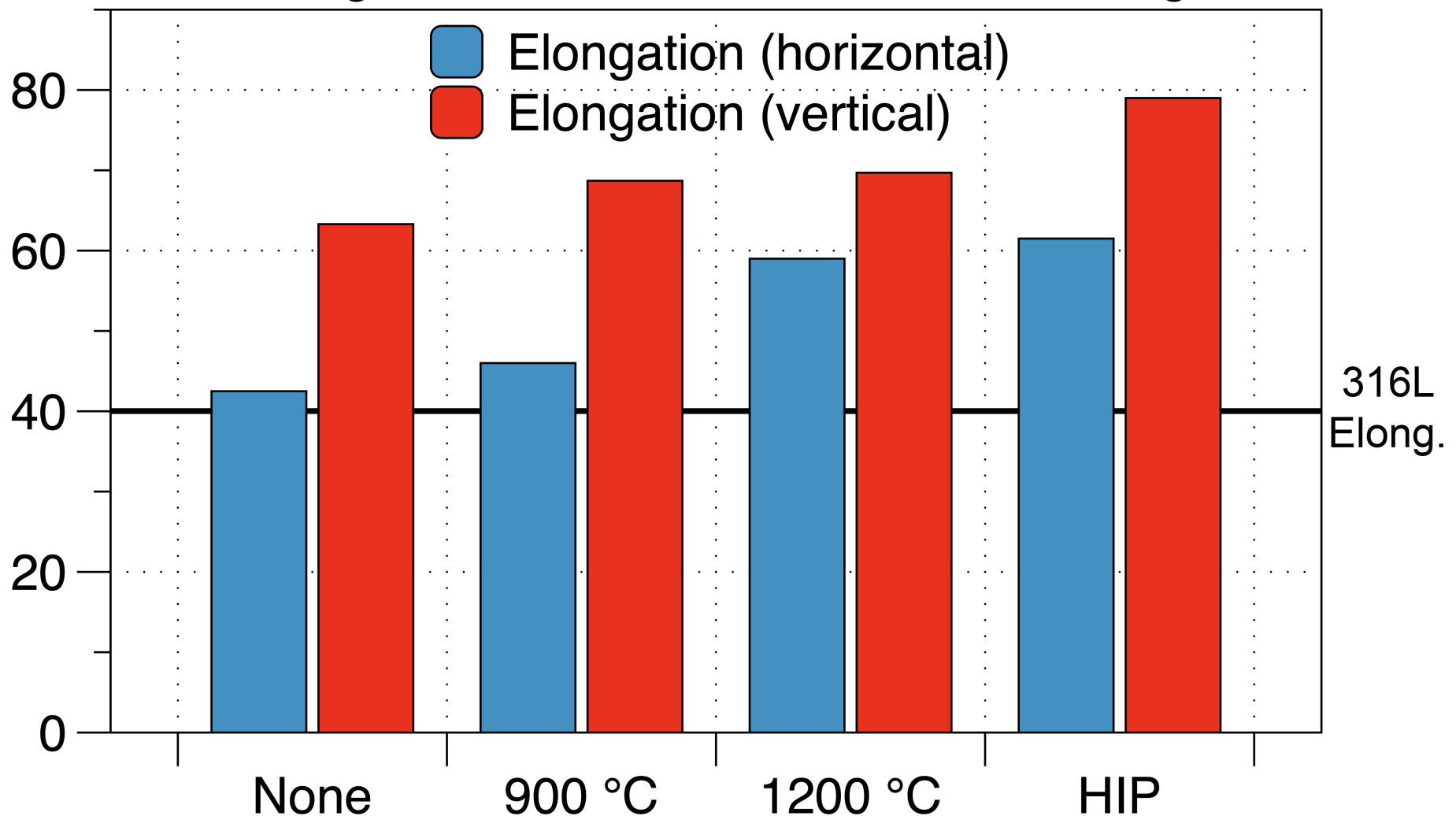


- UTS shows only slight decrease with Temp
- All samples exceed 316L specifications



- Elongation increases with processing temp
- Vertical samples more ductile than horizontal

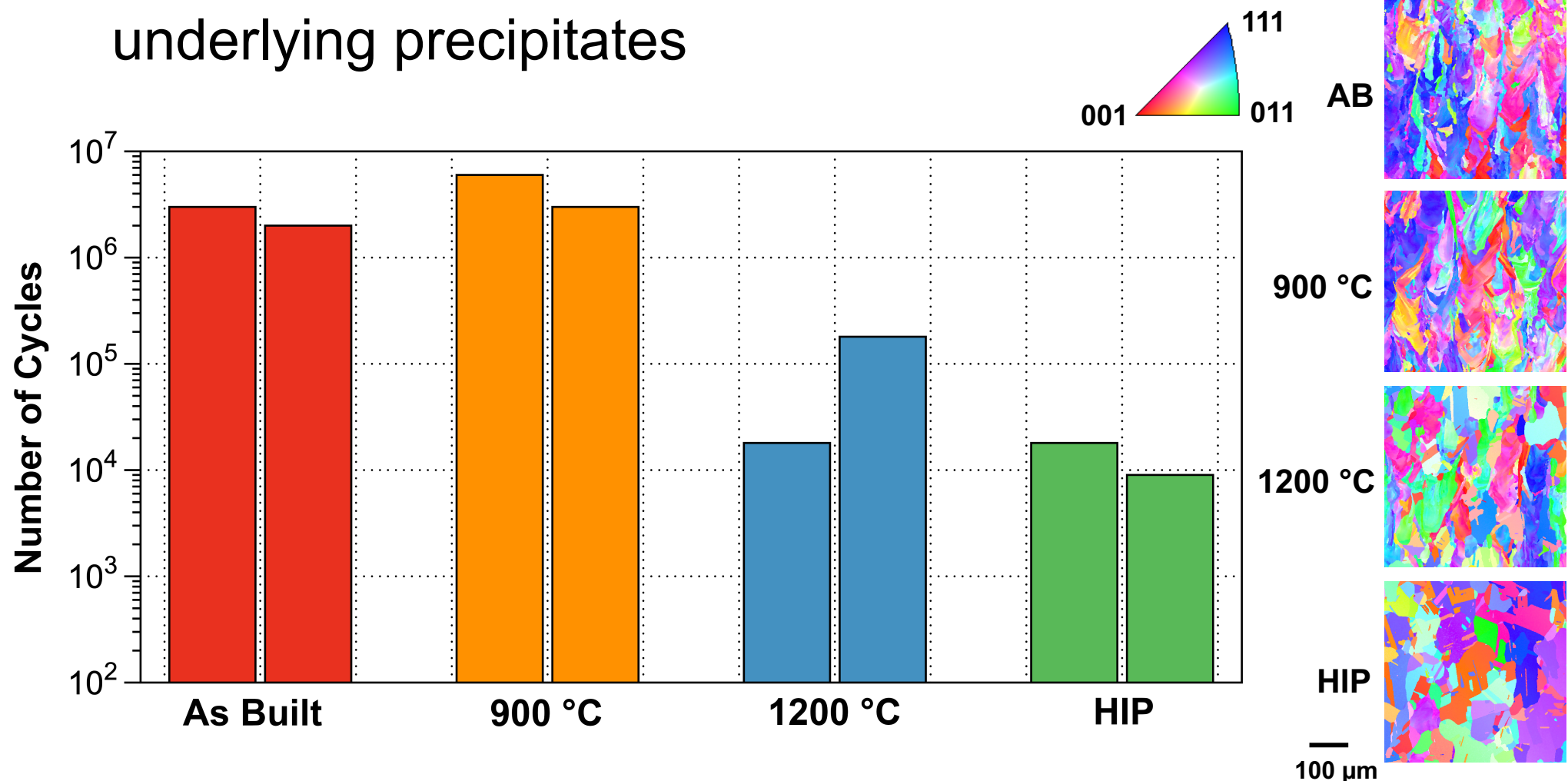
Elongation as Function of Post Processing



Rotate Bending Fatigue

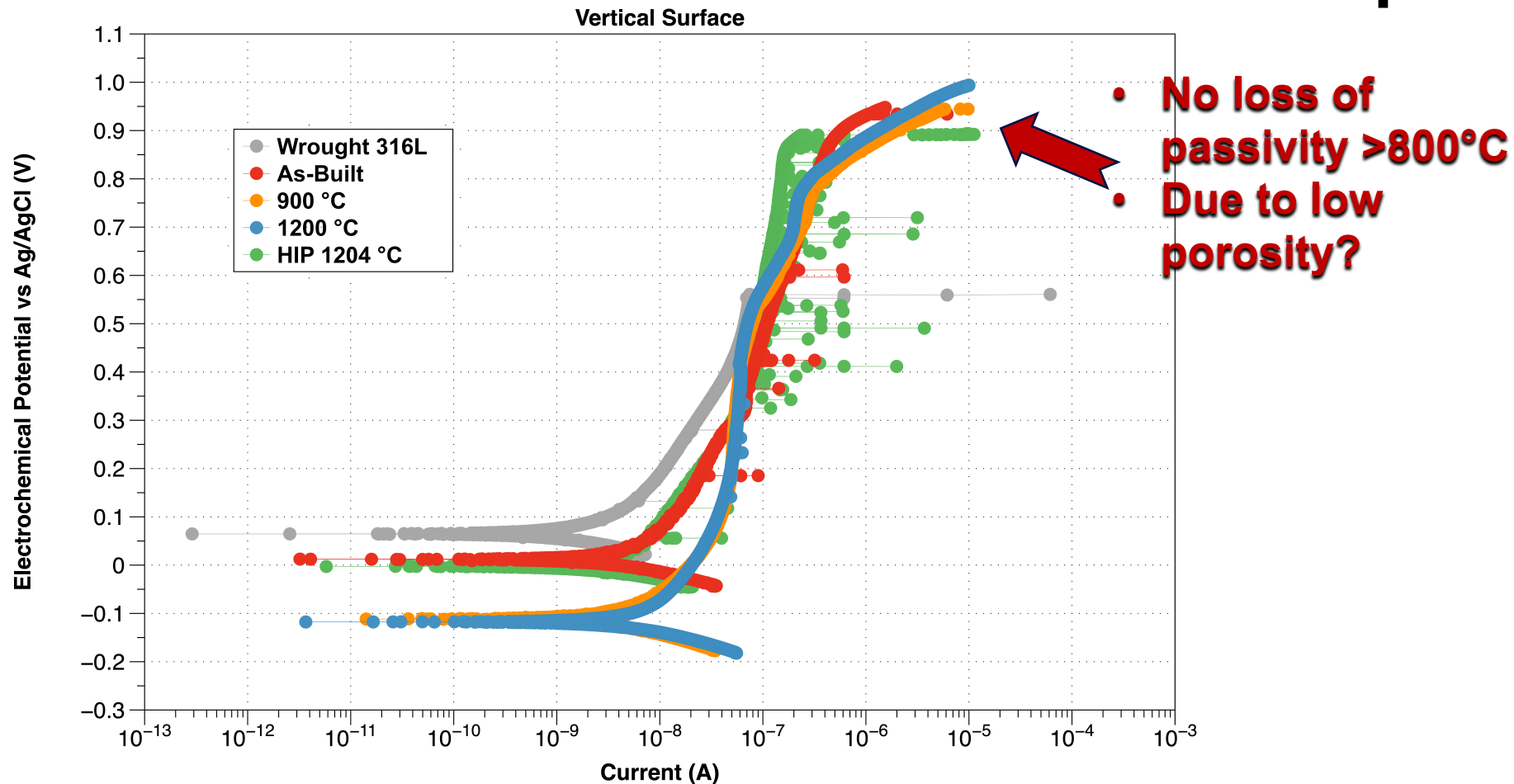
- **Fatigue life degrades with recrystallization**

- Vertical sample orientation (along BD)
- Evolution from vertical grains to equiaxed grains with underlying precipitates

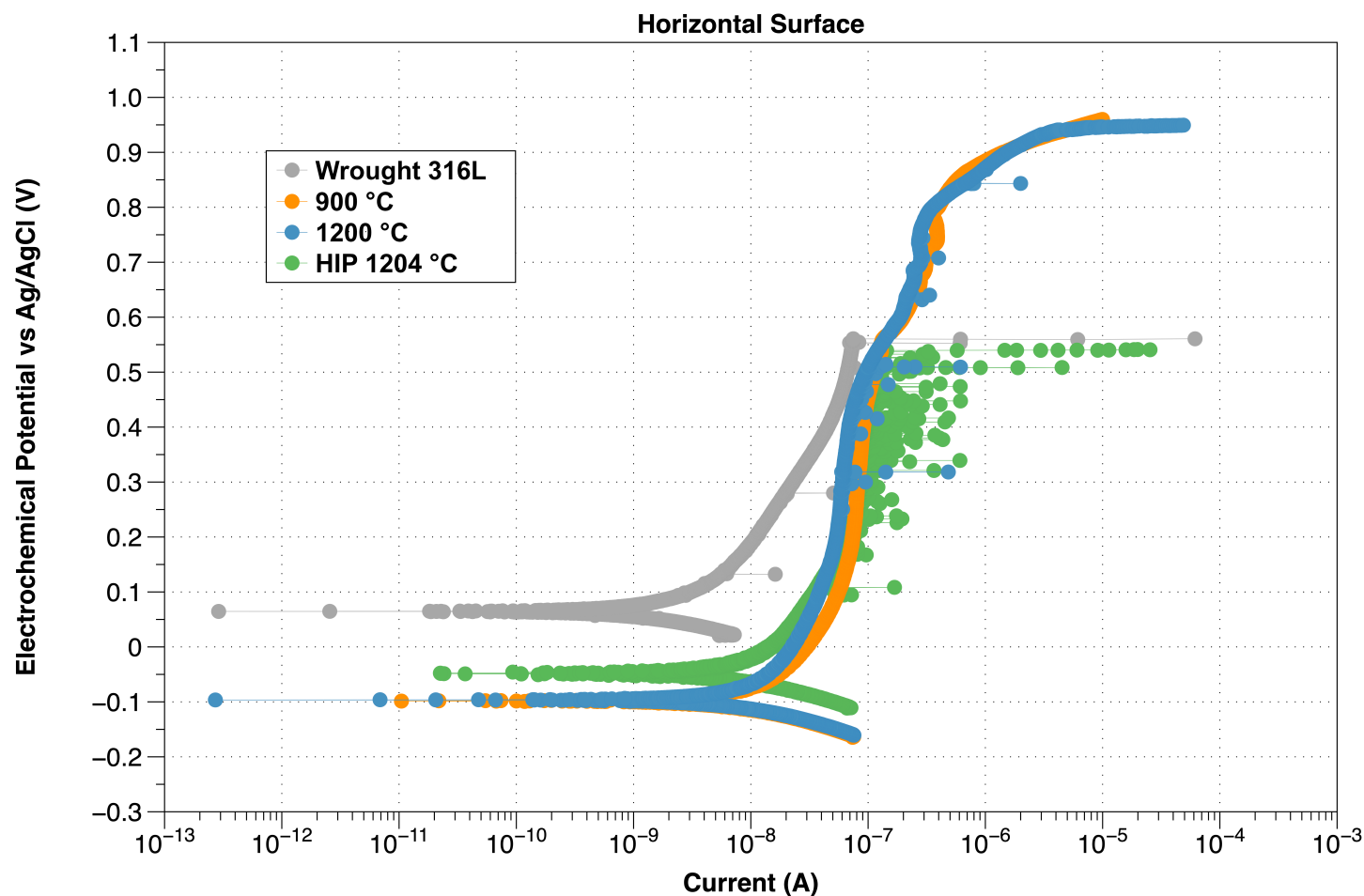


Corrosion at Transverse Surface

- AM has better passivity than the wrought 316L
- HIP sample has transients in passivation regime
 - reflect individual isolated corrosion events (precipitates?)
- Difference in OCP due to defects/matrix comp?

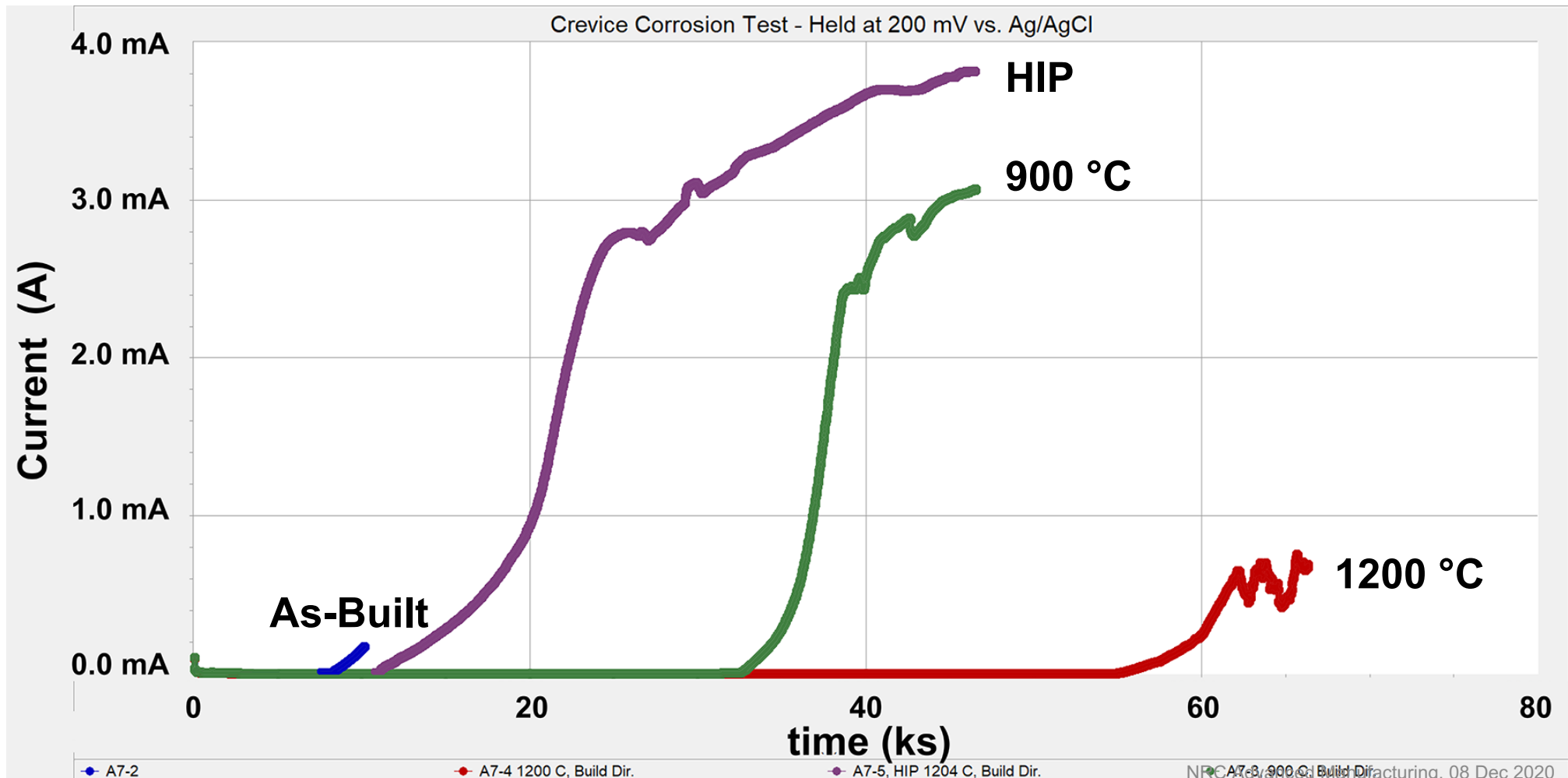


- Generally similar to transverse surface
- More transients observed in HIP sample
 - Leads to earlier breakdown of surface (ppts?)



Onset of Crevice Corrosion

- Delayed onset from AB \Rightarrow 900°C \Rightarrow 1200°C due to coarsening of microstructure scale
- Accelerated CC of HIP condition: precipitates?



- **Corrosion** enhanced at surface due to porosity
- **Three distinct microstructural regimes:**
As-Built < 800 °C < **Annealed** < 1200 °C < **Recrystallized**
- **Microhardness** decreases with increasing temperature
- **Porosity fraction** does not evolve with temperature
 - HIP reduces porosity by ~2/3 by closing LOF pores
- **Yield Strength:** ~3x the 316L specification
 - decreases with processing temperature
- **UTS:** slight decrease due to recrystallization/ppt
- **Elongation:** increases with processing temperature
- **RB Fatigue:** ~100x decrease with recrystallization
- **Corrosion:** HIP-induced precipitation causes increased frequency of transients and rapid onset of crevice corr.
- **Crevice corrosion** resistance improves at 900 °C & 1200 °C
- **All AM structures exceed 316L specifications for YS, UTS, and elongation**

Thank you!